

Effects of trimethylaluminium and tetrakis(ethylmethylamino) hafnium in the early stages of the atomic-layer-deposition of aluminum oxide and hafnium oxide on hydroxylated GaN nanoclusters

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Abstract We calculate the interactions of two atomic layer deposition (ALD) reactants, trimethylaluminium (TMA) and tetrakis(ethylmethylamino) hafnium (TEMAH) with the hydroxylated Ga-face of GaN clusters when aluminum oxide and hafnium oxide, respectively, are being deposited. The GaN clusters are suitable as testbeds for the actual Ga-face on practical GaN nanocrystals of importance not only in electronics but for several other applications in nanotechnology. We find that TMA spontaneously interacts with hydroxylated GaN; however it does not follow the atomic layer deposition reaction path unless there is an excess in potential energy introduced in the clusters at the beginning of the optimization, for instance, using larger bond lengths of various bonds in the initial structures. TEMAH also does not interact with hydroxylated GaN, unless there is an excess in potential energy. The formation of a Ga—N(CH₃)(CH₂CH₃) bond during the ALD of HfO₂ using TEMAH as the reactant without breaking the Hf—N bond could be the key part of the mechanism behind the formation of an interface layer at the HfO₂/GaN interface.

Keywords DFT · Dissociation · Interface layer · Nanoclusters · Wide bandgap semiconductor

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Introduction

Nanotechnology provides excellent opportunities to develop all sorts of devices with tuned properties not only for electronics but practically for all areas of science and engineering. A straightforward way to reach these goals is to prepare nanoclusters whose properties depend strongly on their size and shape, in addition to their intrinsic characteristics of their component atoms and interactions. Currently, enhancement-mode (E-mode) III-V based devices are needed for low-loss high-power RF switching along with transmitter/receiver switches for global positioning system (GPS) receivers in mobile phones [1]. The achievement of E-mode III-V based devices, as high electron mobility transistors (HEMTs), is dependent on the ability to control the threshold voltage (V_{th}); one current method for this is using high-k metal gate stacks (that include high-k dielectrics such as AlO₂ and/or HfO₂), which can vary the work function of the gate metal and therefore the threshold voltage [2–4]. Previous work reported the formation of an interfacial layer at the atomic-layer-deposited (ALD) HfO₂–GaN interface [5, 6], while no interfacial layer has been found for ALD of Al₂O₃ on GaN [7, 8]. The formation of an interfacial layer at the semiconductor/dielectric interface can cause detrimental effects on the performance and the reproducibility of the device along with changes in the threshold voltage of the device. A better understanding of the mechanism responsible for the formation of an interfacial layer during the atomic layer deposition of HfO₂ on GaN would give us insights into determining the removal or the non-formation of the interfacial layer.

In this paper, the interactions of trimethylaluminium (TMA) and tetrakis(ethylmethylamino) hafnium (TEMAH) with the hydroxylated Ga-face of GaN clusters are studied using density functional theory (DFT) with the intent to improve the development of E-mode III-V based devices and to investigate the features of these devices as they are scaled down to nanosizes.

The first step in many atomic layer deposition processes is to introduce H₂O or another oxygen source to oxidize the surface of the sample; an example of this is the use of deionized water as a source of oxidizing agent [9, 10]. In order to fully understand the mechanisms of the atomic layer deposition process of HfO₂ and Al₂O₃, the oxidation of GaN surface must first be understood. Theoretical calculations of the interactions of O₂ and/or H₂O with GaN (0001) have previously been reported using density functional theory (DFT) within the generalized gradient approximation (GGA) [11, 12], and using hybrid functionals [13–15]. TMA and TEMAH are commonly used as precursors during the atomic layer deposition process of Al₂O₃ and HfO₂, respectively. The authors believe, forming individual devices with small nano-clusters could expand their applications beyond photonics and electronics areas such as catalysis, sensing [16], and biomedical applications. Recently, GaN nanoclusters have been made by using the sequential ion implantation on a dielectric matrix [17]. Another method to form GaN nanoclusters is to rinse the sample in a solution of a donor stabilized gallium-triazid [18]. These may be considered nanoclusters of GaN adsorbed over the surface of a thin film. Materials made from or containing GaN have been extensively studied since the early 1990s theoretically using *ab initio* electronic structure methods for GaN clusters [19–28]. The theoretical studies of structural, dielectric, phonon dispersions and density of states for wurzite and/or zinc-blende GaN along with the stoichiometry and structure of the (0001) surface of GaN have been reported [19, 21, 22]. The structural and electronic properties of GaN clusters have been previously modeled [23–25, 28].

Methodology

We use density functional theory [29, 30] with the B3PW91, which is a hybrid functional that uses the Becke exchange [31] combined with a Hartree-Fock [32–34] component and the correlation functional of Perdew-Wang [35–39]. This functional has been used successfully in a variety of applications involving related clusters of atoms [26, 40–42] and other applications involving molecular electronics devices [16, 43]. It has also been used in applications involving metallic clusters [44–50]. The basis set and effective core potential used together with the hybrid functional is the LANL2DZ [51–53], which has also been proved to yield acceptable theoretical predictions [16, 43, 54–57]. Optimizations are carried out with the Bery method [58, 59]. All of the optimized stable structures are checked to be local minima by calculating the Hessian matrix, i.e., the second derivative of the energy with respect to all coordinates. When we find that a structure yields negative Hessian eigenvalues, the geometry is modified accordingly in order to reach a stable structure. All of the DFT calculations are performed with the program GAUSSIAN-09 [60].

The reaction mechanisms of two atomic layer deposition reactants with hydroxylated GaN(0001) surface are studied using hexagonal wurzite crystal geometries that resemble bulk GaN with $a=3.189 \text{ \AA}$ and $c=5.186 \text{ \AA}$ [61]. Hexagonal wurzite crystal geometries are considered the most thermodynamically stable structures for GaN. The Ga surface of GaN has each Ga bonded to three N's and maintains one dangling bond perpendicular to the surface. The GaN clusters are hydroxylated using 1, 2 or 3 –OH groups bonded to the Ga of the surface. Three main surface structures are used to model hydroxylated bulk Ga-face GaN, Ga-centered, N-centered, and hollow-centered surface structures. Two different atomic layer deposition (ALD) reactants are used: trimethylaluminium (TMA) and tetrakis(ethylmethylamino)hafnium (TEMAH) (Fig. 1). TMA is used as the reactant in ALD when Al₂O₃ is being deposited, while TEMAH is used when HfO₂ is being deposited.

Calculations for the hydroxylated Ga-centered, N-centered, and hollow-centered surface clusters (Fig. 2) under the presence of an ALD reactants are calculated with varying multiplicities. In some cases atoms of the reactants and of the –OH group are shifted to induce interactions. Once a stable interaction between the ALD reactant and the hydroxylated GaN cluster occurs, further geometry optimizations are completed to investigate the formation of the Ga–O–Hf and Ga–O–Al bonds.

In the following two sections we discuss the cases when trimethylaluminium (TMA) is used as reactant when Al₂O₃ is deposited (ALD) on GaN and when tetrakis(ethylmethylamino)hafnium (TEMAH) is used as reactant when HfO₂ is deposited on GaN; therefore, we will focus on the interaction of reactants with the surface whereby a layer is formed in between the HfO₂ and the surface.

Trimethylaluminium reactant for aluminum oxide deposition on gallium nitride

The early steps of the deposition of Al₂O₃ on GaN using TMA is studied by finding stable structures of the complexes obtained by placing the TMA near the hydroxylated variants

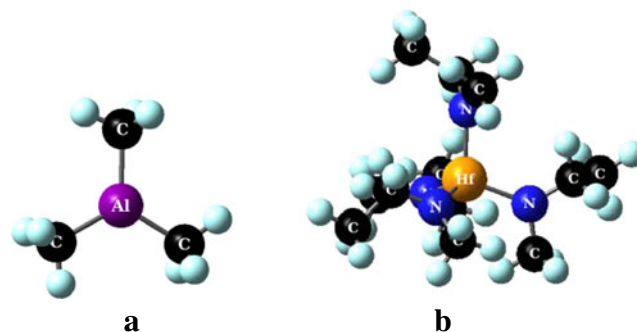
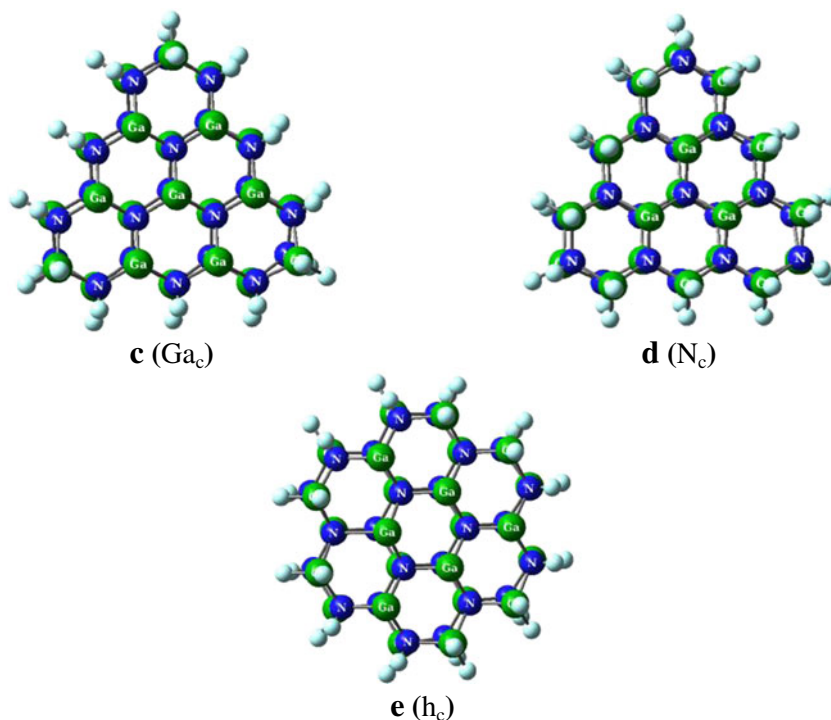


Fig. 1 **a** Trimethylaluminium (TMA) and **b** tetrakis(ethylmethylamino)hafnium (TEMAH). H: white, C: black, Al: purple, Hf: yellow

Fig. 2 Top views of optimized GaN clusters used as surface reactive centers of a GaN surface **c**: Ga-centered, **d**: N-centered and **e**: hollow centered. Ga: green, N: blue



of GaN. Three GaN models are used: the Ga-centered (Ga_c), N-centered (N_c) and hollow-centered (h_c). The ALD of Al_2O_3 on GaN is then represented in these optimized complexes (Fig. 3). The total energies are shown in Table 1 for all these complexes; only geometries tested that yield stable complexes are reported.

For the Ga_c cluster, the most common stable interaction between the TMA and $\text{Ga}_c(\text{OH})$ leads to the bonding of the TMA to the OH group without breaking any bonds in the complex, $\text{Ga}_c(\text{OH})+\text{TMA}$ (**f**, **g**, **j**, **k**). However, the most stable interaction yields the formation of a CH_4 and an O—Al bond (**h**) starting from a TMA with a stressed Al—CH₃ bond of 2.63 Å rather than typical Al—C bond length in TMA of 1.99 Å [62], while the group OH length is 1.16 Å rather than the typical length of 0.98 Å.

These shifts in bond lengths are necessary to form a GaO—Al bond and a CH_4 , which are believed to be the result of TMA reacting with an —OH group [63, 64]. Without shifting any atoms in the complex, the most stable interaction between TMA and $\text{Ga}_c(\text{OH})$ has a final relative energy (E_R) of 0.78 eV (**f**).

The above procedure is repeated for the single TMA molecule placed now over the double and triple hydroxylated Ga_c cluster, $\text{Ga}_c(\text{OH})_2$ and $\text{Ga}_c(\text{OH})_3$, respectively, in several orientations and distances from the surface and optimized (**l-s**). The most stable interaction of TMA with either $\text{Ga}_c(\text{OH})_2$ (**l**) or $\text{Ga}_c(\text{OH})_3$ (**r**) leads to a bonding of the TMA to one of the —OH groups without breaking any bonds in the structure. However, other optimized geometries also reached similar structures with slightly higher energies (Table 1). The formation of CH_4 does not occur for any geometry tested, including

non-stable geometries. None of the atoms are shifted to induce a reaction in the $\text{Ga}_c(\text{OH})_2$ (**l-o**) or $\text{Ga}_c(\text{OH})_3$ (**p-s**) geometries.

Then, Two TMA molecules are placed over the double and triple hydroxylated clusters, in several orientations and distances from the surface and then optimized (**t-y**). The most stable interaction of TMA with either $\text{Ga}_c(\text{OH})_2$ (**t**) or $\text{Ga}_c(\text{OH})_3$ (**x**) yields a bonding of each of the TMA molecules to one of the —OH groups without breaking any bonds in the complex. Likewise other optimized geometries also have similar structures with slightly higher energies (Table 1), and the formation of CH_4 does not occur for any geometry tested, including non-stable geometries. Atoms are shifted to induce a reaction in the $\text{Ga}_c(\text{OH})_2+2\text{TMA}$ (**u**) complex; however, shifting does not yield the formation of a CH_4 , in fact, the geometry failed to converge.

For the N-centered GaN cluster, N_c , a single TMA is placed on a single and a triple hydroxylated $\text{N}_c(\text{OH})$ and $\text{N}_c(\text{OH})_3$, respectively, in several orientations and distances from the surface. These complexes are then optimized (**z** and **aa**) and only the stable structures are shown. The only geometry that yields a stable bonded structure is $\text{N}_c(\text{OH})_3$ (**aa**); this interaction leads to the bonding of TMA to one of the —OH groups without breaking any bonds in the TMA or the GaN cluster. The formation of CH_4 does not occur for any geometry tested, including non-stable geometries, which are not shown.

Similarly, for the hollow-centered GaN, h_c , a single TMA molecule is placed over the singly hydroxylated h_c cluster, $\text{h}_c(\text{OH})$, in several orientations and distances from the surface and optimized (**ab-ad**). The most stable interaction yields the formation of an O—Al bond and a CH_4 (**ad**) from a TMA

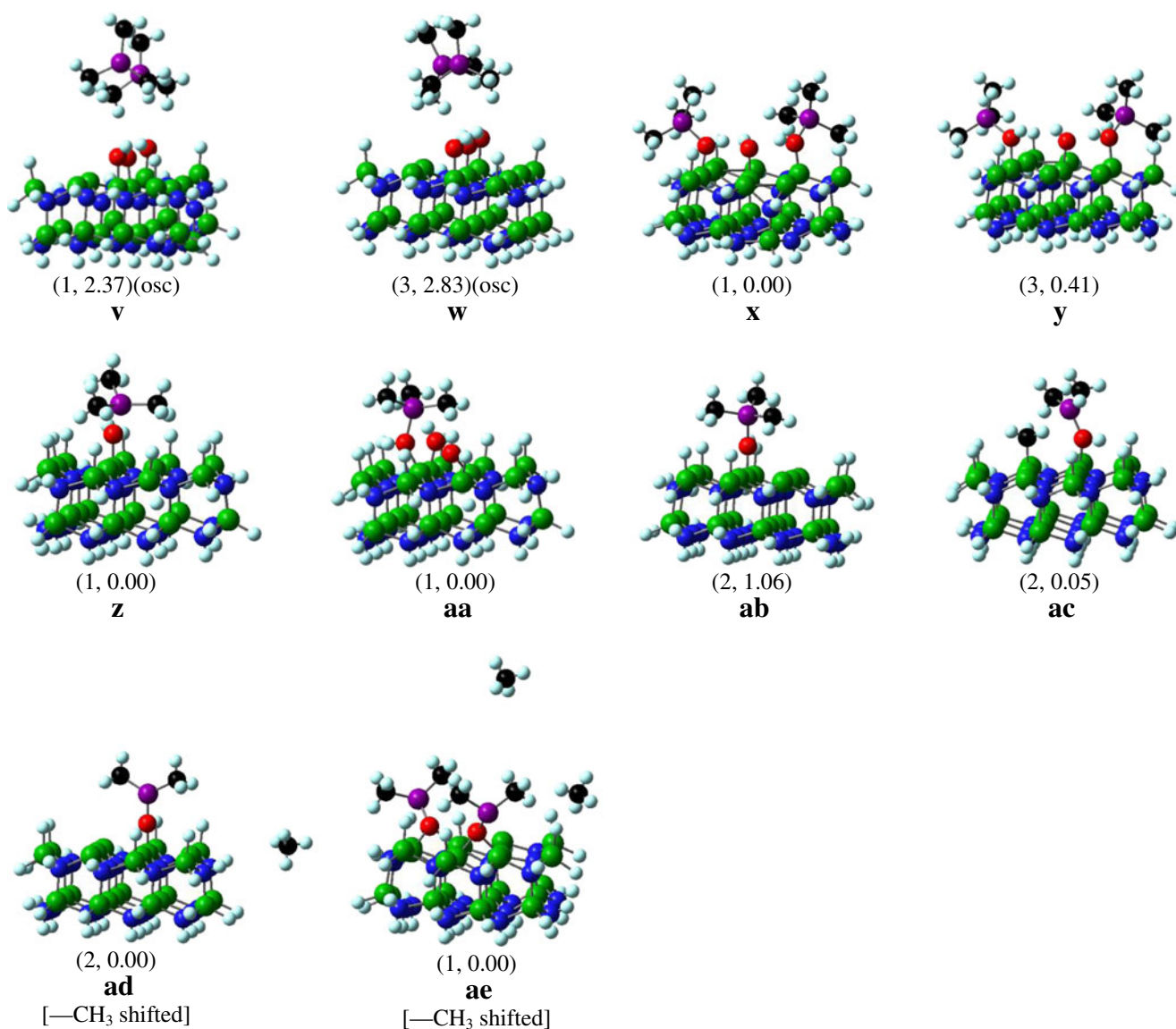


Fig. 3 (continued)

molecule that had been modified by breaking the Al—CH₃ bond. It is important to note the formation of a Ga—OH—Al bond accompanied by the formation of a Ga—CH₃ bond in **ac**. (Heretofore, we will be using “—” to refer to the specific bond under analysis). While the overall lowest energy is not obtained due to a sharp local minimum in the potential surface which causes the structure to oscillate in energy, the lowest energy is near the energy reported.

When two TMA molecules are placed over a double hydroxylated h_c cluster, $h_c(OH)_2$, and optimized (**ae**), the TMA is modified by shifting a CH₃ away from the Al and the only stable interaction of TMA with $h_c(OH)_2$, $h_c(OH)_2+2TMA$ (**ae**), yields the formation of two O—Al bonds, two Ga—O—Ga bonds and two CH₄. This is the first time the formation of an O bonded to two surface Ga's and an Al has taken place within this study; however, due to the lack of another stable

$h_c(OH)_2+2TMA$ complex the only conclusion that can be drawn is that this structure may occur during the deposition process, it is not necessarily the most likely to occur.

The most common stable interactions between the hydroxylated GaN region and one or two TMA molecules generally leads to the formation of a Ga—OH—Al bond with only slight shifting of the bond lengths and angles of the Ga—O bonds, seen in **f**, **g**, **j–o**, **r–u**, **x**, **y**, **aa**, **ab** and **ac**. In the ALD of Al₂O₃ it is believed that O reacts with the Al to form a bond followed by the formation of CH₄. However, in this study it appears that the H prefers to remain bonded to the O rather than the CH₃. This could be due to the fact that under deposition conditions the reactants and substrate are heated to 250 °C thus increasing the kinetic energy of the molecules which may allow for the formation of CH₄, as seen in some of the modified TMA molecules in the complexes **h**, **ad** and **ae**. The modification

Table 1 Total optimized energies (E) for hydroxylated Ga_c , N_c and h_c GaN clusters interacting with 1 or 2 TMA, the number of $-OH$ groups bonded to the GaN cluster, the number of TMA, energies of the initial (E_{RI}) and final (E_R) structures relative to the most stable conformation, the

multiplicity (m) used, the initial (d_i) and final distances (d_f) from the O to the nearest Al, the formation of a CH_4 is determined and the shortest Ga–N bond length (d_{Ga-N}) for the Ga that is taking part in the reaction and the N that is part of the active region GaN

Structures (Fig. 3)	TMA	-OH	m	E (Ha)	E_{RI} (eV)	E_R (eV)	d_i (Å)	d_f (Å)	CH_4 formation?	d_{Ga-N} (Å)	
Ga_c	f	1	1	1	-1471.13034	2.74	0.78	3.00	1.99	–	1.93
	g			3	-1471.11951	2.74	1.08	3.00	2.01	–	1.93
	h			1	-1471.15905	4.90	0.00	3.00	1.72	Yes	1.95
	i			1	-1471.09484	2.38	1.75	2.50	5.57	–	1.94
	j			1	-1471.13018	1.82	0.79	3.00	1.98	–	1.92
	k			3	-1471.11952	4.99	1.08	2.00	2.01	–	1.93
	l	1	2	2	-1547.02606	5.81	0.00	2.50	1.94	–	1.94
	m			4	-1547.01456	5.81	0.31	2.50	1.94	–	1.94
	n			2	-1546.99949	4.92	0.72	2.00	2.01	–	1.92
	o			4	-1546.98843	4.92	1.02	2.00	2.01	–	1.93
	p	1	3	1	-1622.88044 ^a	4.29	0.60	3.60	4.05	–	1.96
	q			3	-1622.85887	4.29	1.19	3.60	4.13	–	1.95
	r			1	-1622.90255	9.45	0.00	2.50	1.99	–	1.94
	s			3	-1622.89368	9.45	0.24	2.50	1.95	–	1.94
	t	2	2	2	-1668.74865	11.62	0.00	1.50	1.99	–	1.93
	u			2	-1668.73673 ^b	9.94	0.33	1.98	1.99	–	1.93
v	2	3	1	-1744.55794 ^b	7.07	2.47	2.70	4.41	–	1.95	
w			3	-1744.54447 ^b	7.07	2.83	2.70	4.64	–	1.96	
x			1	-1744.64855	4.71	0.00	2.20	1.96	–	1.93	
y			3	-1744.63338	4.71	0.41	2.20	1.94	–	1.93	
N_c	z	1	1	1	-1475.81611	3.88	0.00	2.00	2.03	–	1.93
	aa	1	3	1	-1627.57598	11.87	0.00	2.00	1.98	–	1.98
h_c	ab	1	1	2	-1587.84313	5.37	1.06	2.00	2.02	–	1.94
	ac			2	-1587.88016	10.75	0.05	1.45	1.90	–	1.76
	ad			2	-1587.88218	5.51	0.00	1.98	1.73	Yes	1.96
	ae	2	2	1	-1785.58088	15.27	0.00	1.98	1.82	Yes	1.89

^a Imaginary frequencies (cm^{-1}): 94i

^b oscillatory convergence

of TMA yields some of the most stable interactions for hydroxylated Ga_c (**h**) and h_c (**ad**) which form a Ga–O–Al bond and at least one CH_4 , with a binding energy of -52.6 and -42.7 kcal mol^{-1} , respectively. These binding energies are some of the higher values for this type of interaction. The Al–O bond lengths of **h**, **ad**, **ae** (1.76 Å) are in good agreement, compared to 1.8 Å, with previously published data [65, 66]. It is important to mention the formation of a Ga–O–Ga bond where the O is bonded to Al geometry only occurs once throughout this research and is the only stable structure for the given complex.

Stability of the GaN–O–Al bond

The most stable TMA complex with a single, double or triple hydroxylated GaN cluster are used as the initial geometries to determine the stability of the O–Al bond over the GaN

surfaces, Ga–O–Al(CH_3)₂ (Ga–O–dimethylaluminum (DMA)). These geometries are also used to determine the stability of the O–Al bond with other $-OH$ groups or Ga–O bonds on GaN next to or near the O–Al bond (Fig. 4). The total energies are shown in Table 2 for the Ga_c , N_c , and h_c complexes.

Ga-centered GaN clusters (Ga_c) The stability of a single Ga–O–Al(CH_3)₂ geometry is tested using two different geometries bonded to the Ga_c cluster, $Ga_c(O)$ -DMA (**af**, **ag**). The optimization of these geometries does not show deformation or degradation of any bonds in the complex ($Ga_c(O)$ -DMA). The stability of the Ga–O–Al(CH_3)₂ geometry near an $-OH$ group is tested using two geometries, $Ga_c(OH)(O)$ -DMA (**ah**, **ai**). The most stable optimization yields the formation of a Ga–O–Al(CH_3)₂–O–Ga bond (**ah**). Finally the stability of two Ga–

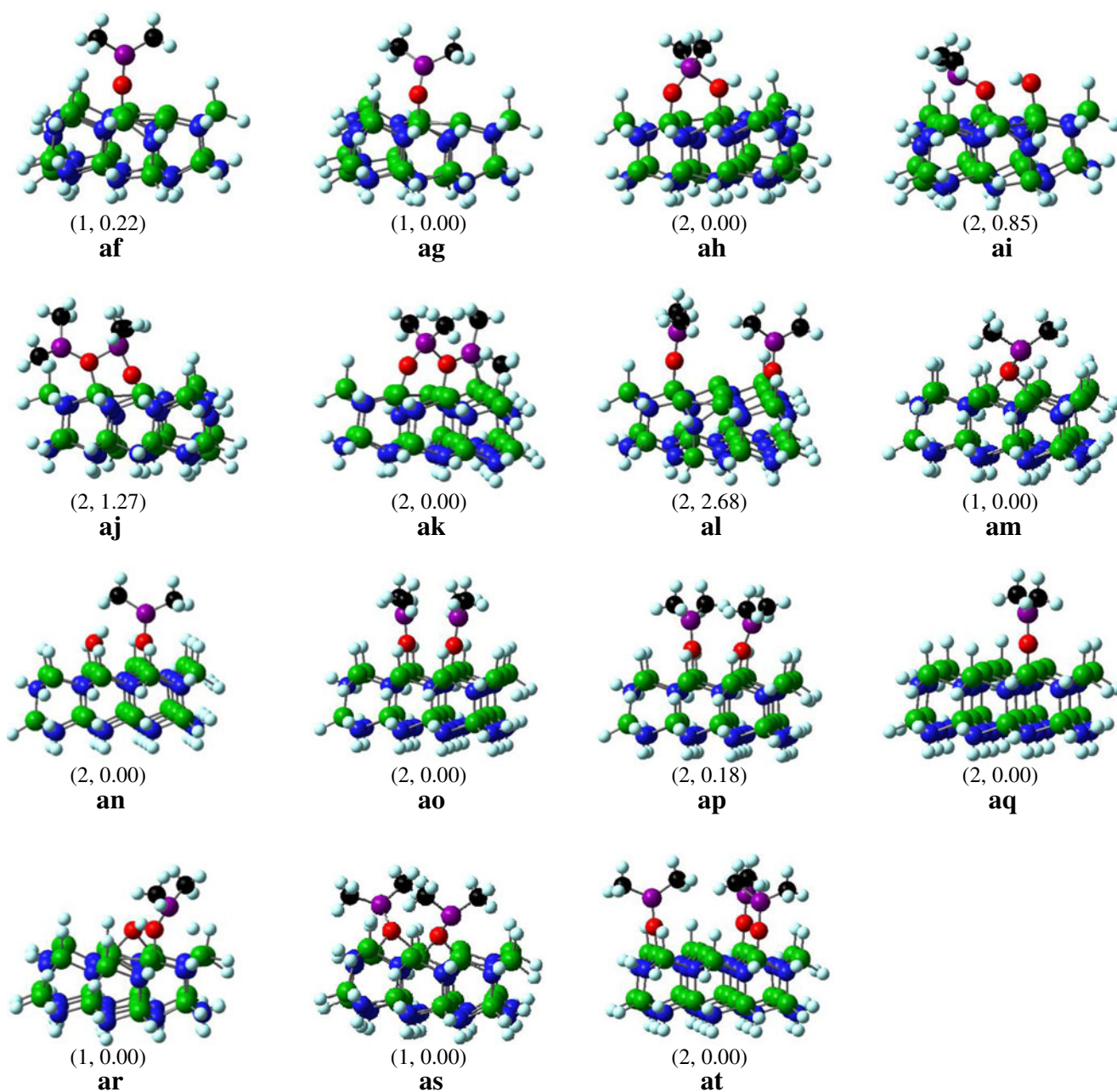


Fig. 4 Stability of 1, 2 or 3 reacted TMA (DMA) with formerly hydroxylated GaN clusters (partially H-passivated), now with 1, 2 or 3 Ga-O bonds, in some cases also in presence of remaining –OH group. The complexes are labeled with (m, E_R), indicating their spin multiplicity (m), and relative energy in eV (E_R) with respect to the most stable conformation, and bold letters; chemical changes with respect to their initial geometries are indicated inside square brackets. For Ga_c GaN clusters, **af-ag**(started from $Ga_c(O)$ -DMA): one Al–O–Ga bond, **ah-ai**($Ga_c(OH)(O)$ -DMA): one Al–O–Ga bond and one –OH group bonded to the GaN and **aj-al**($Ga_c((O)$ -DMA) $_2$),

O–Al(CH₃)₂ geometries on the Ga_c cluster is tested using three separate geometries, $Ga_c((O)$ -DMA) $_2$ (**aj-al**). The overall most stable structure is $Ga_c((O)$ -DMA) $_2$ (**ak**) which forms a Ga–O–Al(CH₃)₂–O–Ga bond and a Ga–O–AlCH₃–Ga bond with a Ga–CH₃ bond at the cluster edge. If there is an –OH group bonded to a surface Ga next to the Ga–O–

two Al–O–Ga bonds have been optimized. For N_c GaN cluster, **am**($N_c(O)$ -DMA): one Al–O–Ga bond, **an**($N_c(OH)(O)$ -DMA): one Al–O–Ga bond and one –OH group bonded to the GaN surface and **ao-ap**($N_c((O)$ -DMA) $_2$): two Al–O–Ga bonds have been optimized. For h_c GaN cluster, **aq**($h_c(O)$ -DMA): one Al–O–Ga bond, **ar**($h_c(OH)(O)$ -DMA): one Al–O–Ga bond and one –OH group bonded to the GaN surface, **as**($h_c((O)$ -DMA) $_2$): two Al–O–Ga bonds, and **at**($h_c((O)$ -DMA) $_3$): three Al–O–Ga bonds have been optimized. O: red

Al(CH₃)₂ then the Al(CH₃)₂ shifts and binds with both O's. However, if the –OH group is not close enough, the Ga–O–Al(CH₃)₂ will remain unchanged.

N-centered GaN cluster (N_c) The stability of a single Ga–O–Al(CH₃)₂ geometry bonded to the N_c cluster is tested, $N_c(O)$ -

Table 2 Total optimized energies (E) to evaluate the stability of reacted TMA (DMA) with formerly hydroxylated Ga_c, N_c and h_c GaN clusters, the number of –OH groups bonded to the GaN cluster, the number of DMA molecules, the number of O's bonded to the GaN cluster, energies of the initial (E_{RI}) and final (E_R) structures relative to the most stable

conformation, the multiplicity (*m*) used, the initial (*d_i*) and final distances (*d_f*) from the O to the nearest Al, the formation of any new or broken bonds is determined and the shortest Ga–N bond length (*d_{Ga-N}*) for the Ga that is taking part in the reaction and the N that is part of the active region GaN

Structure (Fig. 4)	DMA	–OH	O	<i>m</i>	E (Ha)	E _{RI} (eV)	E _R (eV)	<i>d_i</i> (Å)	<i>d_f</i> (Å)	New/broken bonds?	<i>d_{Ga-N}</i> (Å)	
Ga _c	af	1	–	1	1	–1430.65807	0.24	0.22	1.72	1.72	–	1.95
	ag				1	–1430.66615	0.63	0.00	1.72	1.73	–	1.95
	ah	1	1	1	2	–1506.56724	1.09	0.00	1.72	1.78	Yes	1.93
	ai				2	–1506.53610	1.56	0.85	1.72	1.79	–	1.93
	aj	2	–	2	2	–1587.85405	9.11	1.27	1.72	1.81	Yes	1.93
	ak				2	–1587.90084	3.31	0.00	1.72	1.78	Yes	1.94
N _c	al				2	–1587.80229	3.59	2.68	1.72	1.72	–	1.94
	am	1	–	1	1	–1435.38572	1.15	0.00	1.72	1.93	Yes	1.95
	an	1	1	1	2	–1511.22984	1.24	0.00	1.72	1.73	–	1.97
	ao	2	–	2	2	–1592.50047	3.70	0.00	1.72	1.72	–	1.99
	ap				2	–1592.49980	3.48	0.02	1.72	1.72	–	1.99
h _c	aq	1	–	1	1	–1547.38087	0.24	0.00	1.72	1.72	–	1.95
	ar	1	1	1	2	–1623.25510	1.99	0.00	1.72	1.80	–	1.91
	as	2	–	2	1	–1704.57947	2.62	0.00	1.72	1.82	Yes	1.89
	at	3	–	3	2	–1861.68456	0.85	0.00	1.72	1.72	–	1.94

DMA (**am**). The optimization of this geometry yields the formation of a new Ga–O bond, where the O is bonded to two surface Ga's. The stability of the Ga–O–Al(CH₃)₂ geometry near an –OH group is tested in N_c(OH)(O)-DMA (**an**). The optimization does not show the formation or degradation of any bonds in the complex (N_c(OH)(O)-DMA). Finally the stability of two Ga–O–Al(CH₃)₂ geometries on the N_c cluster is tested using two separate geometries, N_c((O)-DMA)₂ (**ao**, **ap**). The optimization does not show the formation or degradation of any bonds only slight shifts in bond lengths and angles of the initial geometry, Table 2.

Hollow-centered GaN clusters (h_c) The stability of a single Ga–O–Al(CH₃)₂ geometry bonded to the h_c cluster is tested, hc(O)-DMA (**aq**). The optimization does not reveal any formation or degradation of bonds of the complex (h_c(O)-DMA). The stability of the Ga–O–Al(CH₃)₂ geometry near an –OH group is tested, hc(OH)(O)-DMA (**ar**), and the optimization of this geometry yields the formation of a Ga–O bond showing an overall geometry of Ga–O–Ga and a floating H while the Ga–O–Al(CH₃)₂ geometry shifted in angle and in bond lengths. The stability of two Ga–O–Al(CH₃)₂ geometries on the hc cluster is tested using one geometry, hc((O)-DMA)₂ (**as**). The optimization yields the formation of a Ga–O–Ga bond where the O is bonded to Al(CH₃)₂. The other Ga–O–Al(CH₃)₂ geometry shifted in angle and in bond lengths. Finally the stability of three Ga–O–Al(CH₃)₂ geometries on the h_c cluster is tested using one geometry, hc((O)-DMA)₃ (**at**). The optimization does not result in the formation or degradation of any bonds only slight shifts in bond lengths and angles, Table 2.

Typically the Ga–O–Al(CH₃)₂ geometry is preserved during optimization if no other molecules are surrounding it (**af**, **ag**, **ai**, **al**, **aq** and **at**). However, in some cases the O or the Al(CH₃)₂ (DMA) shifts to form new bonds (**ah**, **aj**, **ak**, **am**, **ar** and **as**). For an example the Ga–O–Al(CH₃)₂ geometry on the N_c cluster, the O shifts to form a Ga–O–Ga bond without breaking any other bonds (**am**). It is important to note that due to the lack of multiple stable structures it is impossible to know if the structures shown here are the lowest energies achievable. However, some conclusions can be made. First, it is possible to form O–Al(CH₃)₂–O and Ga–O–Ga geometries during ALD of Al₂O₃ on GaN clusters. Second, the Ga–O–Al(CH₃)₂ geometry may interact with unreacted Ga–OH and other Ga–O–Al(CH₃)₂ geometries. Research is currently continuing in this area.

Tetrakis(ethylmethylamino) hafnium for deposition of hafnium oxide on gallium nitride

Interaction of TEMAH with hydroxylated GaN clusters

TEMAH is placed over a hydroxylated Ga_c and h_c GaN clusters to represent the atomic layer deposition (ALD) process of HfO₂ on GaN, Fig. 5. The total energies are shown in Table 3 for the Ga_c and h_c GaN complexes. The interaction of TEMAH with hydroxylated N_c cluster is tested; however, no stable structures are obtained.

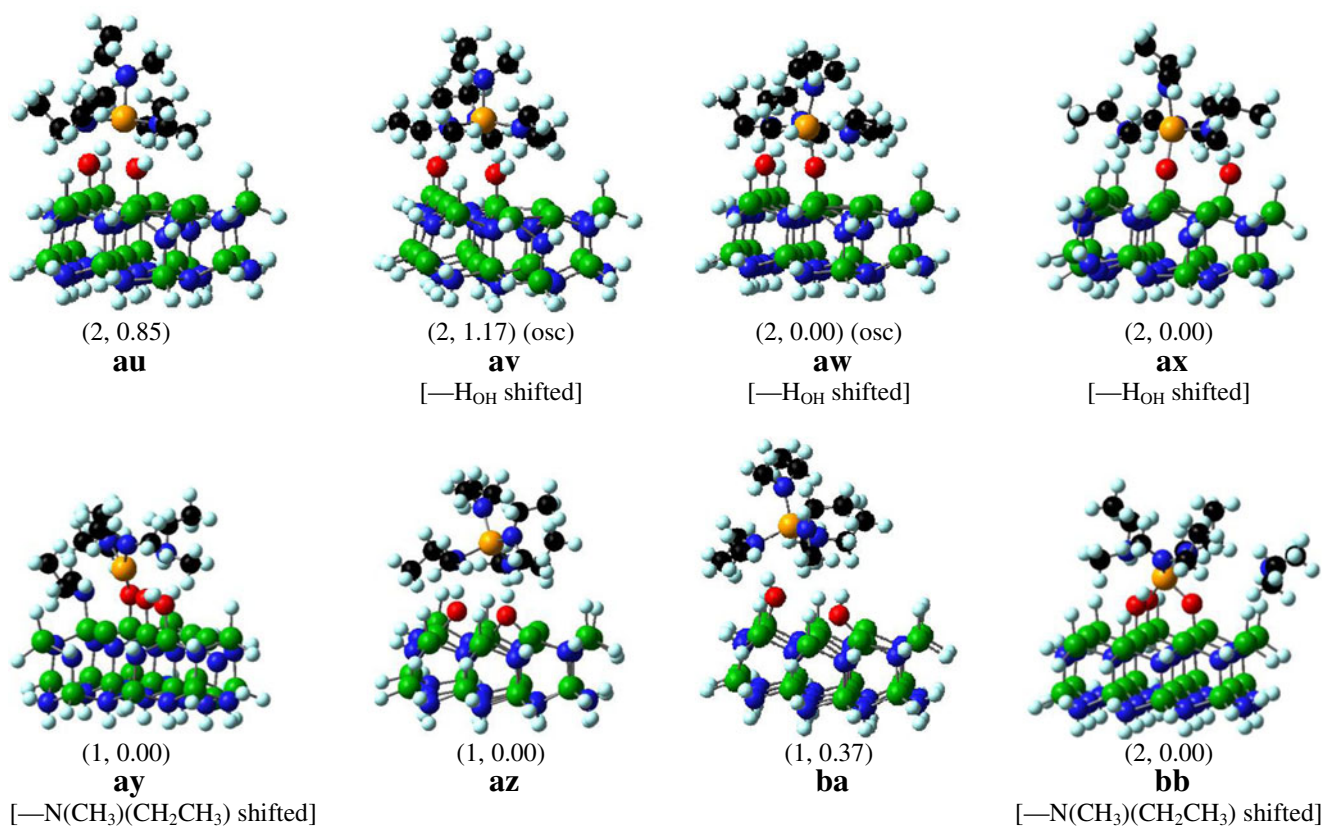


Fig. 5 GaN clusters (partially H-passivated) hydroxylated (with 2 or 3 –OH groups) optimized structures while under the presence of one TEMAH molecule. The complexes are labeled with (m, E_R), indicating their spin multiplicity (m), and relative energy in eV (E_R) with respect to the most stable conformation, and bold letters; chemical changes with respect to their initial geometries are indicated inside square brackets. For Ga_c GaN clusters, **au**(starting from $Ga_c(OH)_2$ +TEMAH): one TEMAH

and two –OH groups, **av–ax**($Ga_c(OH)_2$ +TEMAH): two –OH groups and one TEMAH molecule, re-accommodating H atoms to avoid steric interactions, and **ay**($Ga_c(OH)_3$ +TEMAH): three –OH groups and one TEMAH with a N shifted. For hc GaN cluster, **az–ba**($h_c(OH)_2$ +TEMAH): two –OH groups and one TEMAH molecule and **bb**($h_c(OH)_3$ +TEMAH): three –OH groups and one TEMAH molecule with a N shifted. O: red, osc: failure to converge due to oscillations

Ga-centered GaN clusters (Ga_c) A single TEMAH molecule is placed over a double and triple hydroxylated Ga_c cluster, $Ga_c(OH)_2$ and $Ga_c(OH)_3$, respectively, in several orientations modifying bond lengths of C–H and N–Hf (from Hf– $(N(CH_3)(CH_2CH_3))_3$), and then optimized (**au–ay**). The most stable interaction yields the bonding of the TEMAH to the –OH group which caused Hf– $N(CH_3)(CH_2CH_3)$ bond to break and react with the H from the –OH group forming an H–N bond (**aw–ay**). A shift in either the N–Hf bond or the O–H bond is necessary to form O–Hf and H– $N(CH_3)(CH_2CH_3)$ bonds. Without shifting any atoms in the structure, the most stable structure between TEMAH and $Ga_c(OH)$ resulted in a final relative energy (E_R) of 0.85 eV and no interaction between the hydroxylated surface and TEMAH (**au**).

Hollow-centered GaN clusters (h_c) One TEMAH molecule is placed over the double and triple hydroxylated h_c cluster, $h_c(OH)_2$ and $h_c(OH)_3$ in several orientations modifying bond lengths of C–H and N–Hf, and optimized (**az–bb**). The most stable interaction between TEMAH and the double hydroxylated h_c cluster yields the formation of a Ga–O–Ga

bond without interacting with the TEMAH (**az**). The only stable interaction between TEMAH and the triple hydroxylated h_c cluster yields the bonding of the TEMAH to the –OH group which caused Hf– $N(CH_3)(CH_2CH_3)$ bond to break and react with the H from the –OH group forming an H–N bond (**bb**). A shift in the N–Hf bond is necessary to form O–Hf and H– $N(CH_3)(CH_2CH_3)$ bonds. The formation of H– $N(CH_3)(CH_2CH_3)$ and O–Hf only occurs when either the Hf–N or the O–H bond lengths are increased significantly. Deposition temperatures of the substrate and TEMAH are around 200 °C and 150 °C, respectively. These temperatures correspond to increases in the kinetic energy of the molecules which leads to changes in bond lengths. The research shows that the most stable interaction between TEMAH and hydroxylated GaN clusters results in formation of H– $N(CH_3)(CH_2CH_3)$ and Ga–O–Hf bonds (**aw**, **ax**, **ay**, **bb**), when reaction kinetics allow, i.e., temperatures beyond that of room temperature; such structures have the most favorable binding energy, being for **ay** and **bb** of -61.4 and -56.4 kcal mol $^{-1}$, respectively. The Hf–O bond lengths of 2.02 Å for those molecules that formed N–H bonds, are similar to previously published data, 2.10 Å [67, 68].

Table 3 Total optimized energies (E) for hydroxylated Ga_c and h_c GaN cluster interacting with one TEMAH molecule, the number of $-\text{OH}$ groups bonded to the GaN cluster, the number of TEMAH molecules, energies of the initial (E_{RI}) and final (E_{R}) structure relative to the most stable conformation, the multiplicity (m) used, the initial (d_i) and final

distances (d_f) from the O to the nearest Hf, the formation of a N-H bond is determined and the shortest Ga-N bond length ($d_{\text{Ga-N}}$) for the Ga that is taking part in the reaction and the N that is part of the active region GaN. (Osc) signifies the structure failed to converge due to oscillations

Structure (Fig. 5)	TEMAH	-OH	m	E (Ha)	E_{RI} (eV)	E_{R} (eV)	d_i (Å)	d_f (Å)	N-H formation?	$d_{\text{Ga-N}}$ (Å)
Ga_c	au	1	2	-2169.54182	15.57	0.85	2.40	2.47		1.94
	av		2	-2169.53025 ^a	10.24	1.17	2.49	2.76		1.94
	aw		2	-2169.57318 ^a	4.27	0.00	2.29	2.05	Yes	1.97
	ax		2	-2169.57321	2.34	0.00	2.71	2.05	Yes	1.97
	ay	1	3	-2245.51539	19.55	0.00	2.70	1.99	Yes	1.94
h_c	az	1	2	-2286.29151	15.07	0.00	3.28	3.99		1.89
	ba		1	-2286.27809	3.86	0.37	3.13	3.76		1.92
	bb	1	3	-2362.22714	30.02	0.00	2.00	1.98	Yes	1.96

^a Oscillating convergence

It is important to mention the formation of a Ga-O—Ga bond where the O is bonded to Hf. This geometry only occurs once throughout this research and is the only stable structure for the given structure; therefore no definitive conclusions can be drawn.

Stability of the GaN-O—Hf bond

The most stable TEMAH interaction with a double hydroxylated GaN that matched the generally accepted initial structure O-Hf structure during ALD is used as

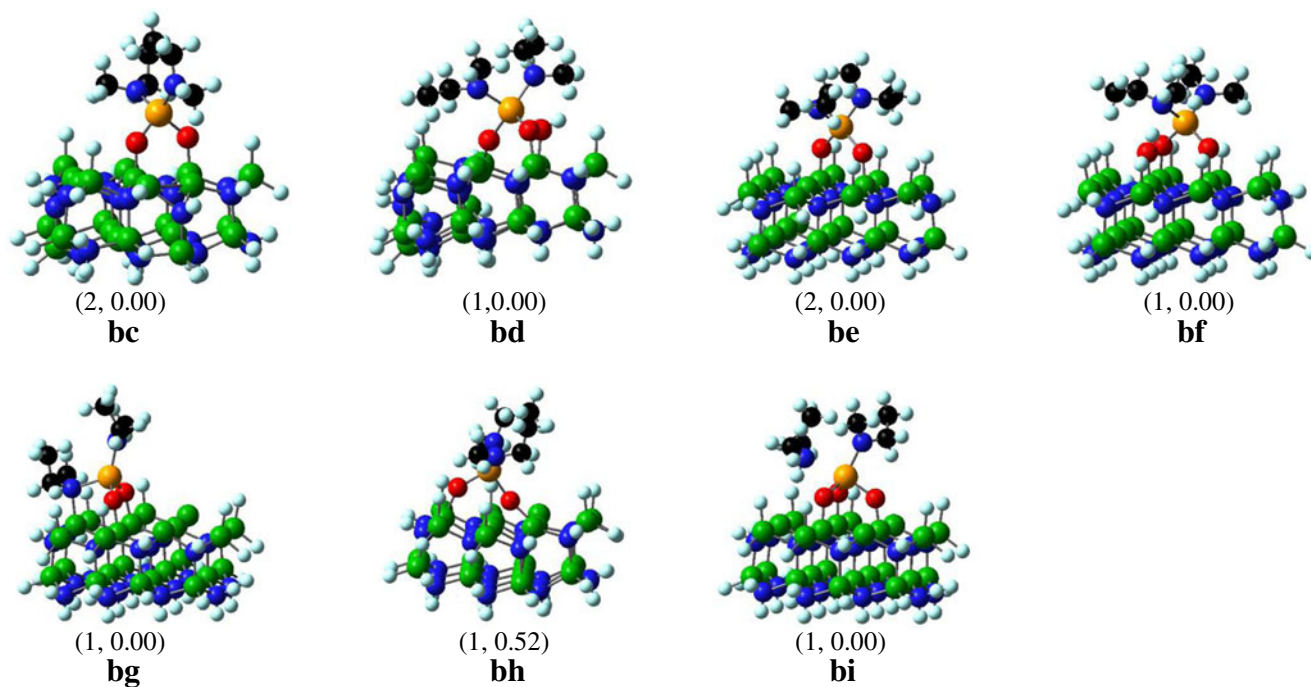


Fig. 6 Stability of one reacted TEMAH (HfAA) with formerly hydroxylated GaN clusters (partially H-passivated), now with 2 or 3 Ga-O bonds. TEMAH reacted on Ga-GaN clusters forming two O-Hf bonds, in some cases under the presence of one remaining $-\text{OH}$ group. The complexes are labeled with (m , E_{R}), indicating their spin multiplicity (m), and relative energy in eV (E_{R}) with respect to the most stable conformation, and bold letters; chemical changes with respect to their initial geometries are indicated inside square brackets. For Ga_c GaN clusters, **bc** (started from $\text{Ga}_c(\text{O})_2\text{-HfAA}$): one Ga-O—Hf(N(CH₃)(CH₂CH₃))₂—O-Ga bond and

bd($\text{Ga}_c(\text{OH})(\text{O})_2\text{-HfAA}$): one Ga-O—Hf(N(CH₃)(CH₂CH₃))₂—O-Ga bond and one $-\text{OH}$ group have been optimized. For N_c GaN clusters, **be**($\text{N}_c(\text{O})_2\text{-HfAA}$): one Ga-O—Hf(N(CH₃)(CH₂CH₃))₂—O-Ga bond and **bf**($\text{N}_c(\text{OH})(\text{O})_2\text{-HfAA}$): one Ga-O—Hf(N(CH₃)(CH₂CH₃))₂—O-Ga bond and one $-\text{OH}$ group have been optimized. For h_c GaN clusters, **bg**-**bh**($\text{h}_c(\text{O})_2\text{-HfAA}$): one Ga-O—Hf(N(CH₃)(CH₂CH₃))₂—O-Ga bond, and **bi**($\text{h}_c(\text{OH})(\text{O})_2\text{-HfAA}$): one Ga-O—Hf(N(CH₃)(CH₂CH₃))₂—O-Ga bond and one $-\text{OH}$ group have been optimized. O: red, Osc: complexes did not reach a minimum due to oscillations

Table 4 Total optimized energies (E) to evaluate the stability of reacted TEMAH(HfAA) with formerly hydroxylated Ga_c , N_c and h_c GaN cluster, the number of $-OH$ groups bonded to the GaN cluster, the number of HfAA molecules, the number of O's bonded to the GaN cluster, energies of the initial (E_{RI}) and final (E_R) structure relative to the most stable

conformation, the multiplicity (m) used, the initial (d_i) and final distances (d_f) from the O to the nearest Hf, the formation of new or broken bonds is determined and the shortest Ga-N bond length (d_{Ga-N}) for the Ga that is taking part in the reaction and the N that is part of the active region GaN

Structure (Fig. 6)	HfAA	-OH	O	m	E (Ha)	E_{RI} (eV)	E_R (eV)	d_i (Å)	d_f (Å)	New/broken bonds?	(d_{Ga-N}) (Å)	
Ga_c	bc	1	–	2	2	-1820.82067	2.78	0.00	2.00	1.94	–	1.94
	bd	1	1	2	1	-1896.70358	5.60	0.00	2.00	1.96	–	1.90
N_c	be	1	–	2	2	-1825.51620	5.08	0.00	2.00	1.95	–	1.96
	bf	1	1	2	1	-1901.39807	7.10	0.00	2.00	1.93	–	1.95
h_c	bg	1	–	2	1	-1937.55161	4.72	0.00	2.00	1.90	Yes	1.92
	bh	1	–	2	1	-1937.53263	5.32	0.52	2.00	1.95	–	1.92
	bi	1	1	2	1	-2013.45498	4.18	0.00	2.00	1.97	Yes	1.96

the initial geometry to determine the stability of the O—Hf bond, over the hydroxylated GaN surfaces. Other geometries are also used to determine the stability of the O—Hf bond with other $-OH$ groups or Ga—O bonds on GaN next to or near the O—Hf bond, Fig. 6. All complexes are optimized and the total energies are shown in Table 4 for the Ga_c , N_c , and h_c complexes.

Ga-centered GaN clusters (Ga_c) The stability of a single Hf— $(N(CH_3)(CH_2CH_3))_2$ (hafnium alkylamide (HfAA)) geometry, where the Hf is bonded to two O's, is tested with several geometries having the O atoms bonded to the Ga_c cluster; however the only complex that reaches a stable structure is $Ga_c(O)_2$ -HfAA (**bc**). The geometry optimization of this complex does not show the formation or degradation of any bonds but only shows the shifting in bond angles and lengths of the complex (**bc**). The stability of the Hf— $(N(CH_3)(CH_2CH_3))_2$

geometry near an $-OH$ group is tested, $Ga_c(OH)(O)_2$ -HfAA (**bd**), and the most stable optimization yields the deformation and thus the upward bending of the GaN cluster (**bd**).

N-centered GaN clusters (N_c) While the interaction between hydroxylated N_c and TEMAH does not result in any stable structures, the stability of a single Hf— $(N(CH_3)(CH_2CH_3))_2$ geometry where the Hf is bonded to two O's is tested with the O's bonded to the N_c cluster, $N_c(O)_2$ -HfAA (**be**). The optimization of this geometry does not show the formation or degradation of any bonds only shifts in bond angles and lengths of the complex $N_c(O)_2$ -HfAA. The stability of the Hf— $(N(CH_3)(CH_2CH_3))_2$ geometry near an $-OH$ group is tested, $N_c(OH)(O)_2$ -HfAA (**bf**), and again the optimization of this geometry does not show the formation or degradation of any bonds only shifts in bond angles and lengths of the complex $N_c(OH)(O)_2$ -HfAA (Table 4).

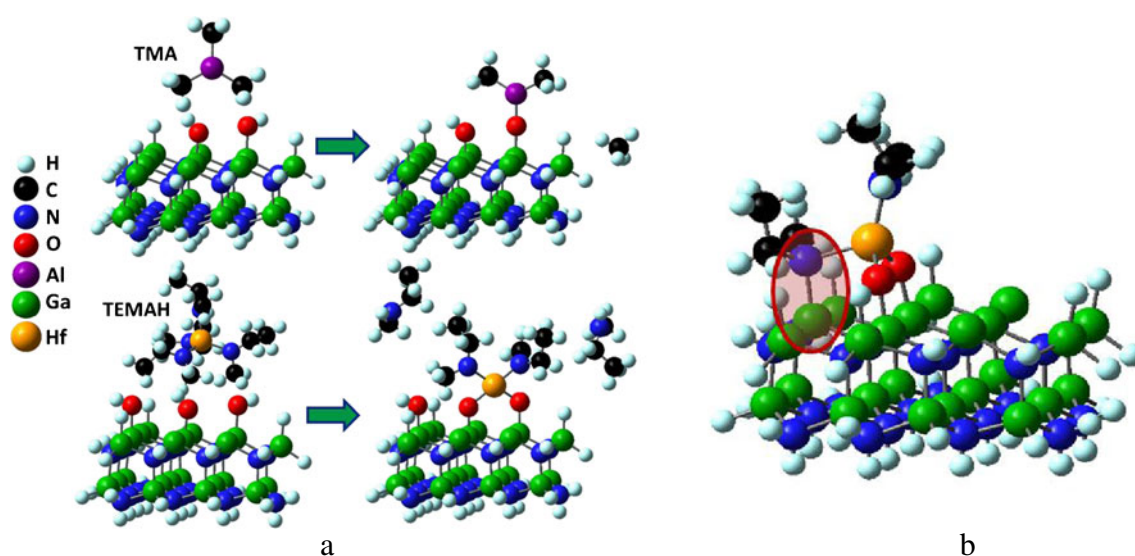


Fig. 7 **a** Interactions of TMA and TEMAH with hydroxylated GaN clusters leading to the formation of CH_4 , which is expected during the ALD process. **b** Formation of Ga— $N(CH_3)(CH_2CH_3)$ bond during deposition of HfO_2 using TEMAH forming the interfacial oxide during deposition of HfO_2 using ALD on GaN

Hollow-centered GaN clusters (h_c) The stability of a single Hf—N(CH₃)(CH₂CH₃)₂ geometry where the Hf is bonded to two O's is tested using two geometries with the O's bonded to the h_c cluster, $h_c(O)_2$ -HfAA (**bg**, **bh**). The optimization of this geometry, given the O's are attached to neighboring Ga surface atoms, yields the formation of a Ga—N(CH₃)(CH₂CH₃) bond without breaking any bonds (**bg**) in the complex $h_c(O)_2$ -HfAA. This interaction may be the mechanism behind the formation of interfacial layer during the ALD of HfO₂ on GaN. The optimization of a similar geometry with the O's separated by four Ga—N bond lengths does not show formation or degradation of any bonds, only shifts in bond angles and lengths (**bh**) of the complex $h_c(O)_2$ -HfAA. The stability of the Hf—N(CH₃)(CH₂CH₃)₂ geometry near an —OH group is tested, $h_c(OH)(O)_2$ -HfAA (**bi**). The most stable optimization, results in the formation of a H—N(CH₃)(CH₂CH₃) bond and an additional O—Hf bond (**bi**). It is important to note the formation of a stable Ga—N(CH₃)(CH₂CH₃) bond (**bg**). While this geometry is the most stable, the **bh** is at only 0.52 eV above. Since the Ga—N(CH₃)(CH₂CH₃) bond is only present in the most stable structure, this bonded structure may form during ALD of HfO₂ on GaN and may be the cause of the interfacial layer formation as no such structure is found in this research of TMA interacting with hydroxylated GaN clusters for fully optimized and stable structures.

It is important to note the formation of a stable Ga—N(CH₃)(CH₂CH₃) bond (**bg**). While this geometry is the most stable, the **bh** is at only 0.52 eV above. Since the Ga—N(CH₃)(CH₂CH₃) bond is only present in the most stable structure, this bonded structure may form during ALD of HfO₂ on GaN and may be the cause of the interfacial layer formation as no such structure is found in this research of TMA interacting with hydroxylated GaN clusters for fully optimized and stable structures.

Conclusions

Through this research, a better understanding of the interactions of TMA and TEMAH with hydroxylated GaN clusters during atomic layer deposition is achieved. TMA spontaneously interacts with hydroxylated GaN; however, the interaction does not result in the formation of a CH₄ which is expected during the ALD process. An increase in kinetic energy is needed to form methane and follow the expected Al₂O₃ ALD process. For the structures that formed a methane molecule, the O—Al bond lengths are similar to previously published works including Al₂O₃. In all other cases the most stable optimized geometries results in the TMA interacting with the hydroxylated GaN to form a Ga—O—Al bond. In one optimized structure, the formation of a Ga—CH₃ bond did occur. Similar to the interaction of TMA with hydroxylated GaN, TEMAH requires an increase in kinetic energy to follow the expected HfO₂ ALD process, e.g.,

the formation of a H—N bond. For the structures that formed a H—N bond, the O—Hf bond lengths are similar to previously published works including HfO₂. It is discovered that the formation of a Ga—N(CH₃)(CH₂CH₃) bond can form during the deposition of HfO₂ using ALD and TEMAH as the reactant without breaking the Hf—N bond; this finding and the general procedure are schematically resumed in Fig. 7. The formation of a Ga—N(CH₃)(CH₂CH₃) bond is significant because with the introduction of water into the complex the methyl and ethyl methyl groups may react to form a Ga—N—O bond which is believed to be the interfacial oxide found during deposition of HfO₂ using ALD on GaN.

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